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Characterization of collagen/lipid nanoparticle-curcumin cryostructurates for wound healing applications

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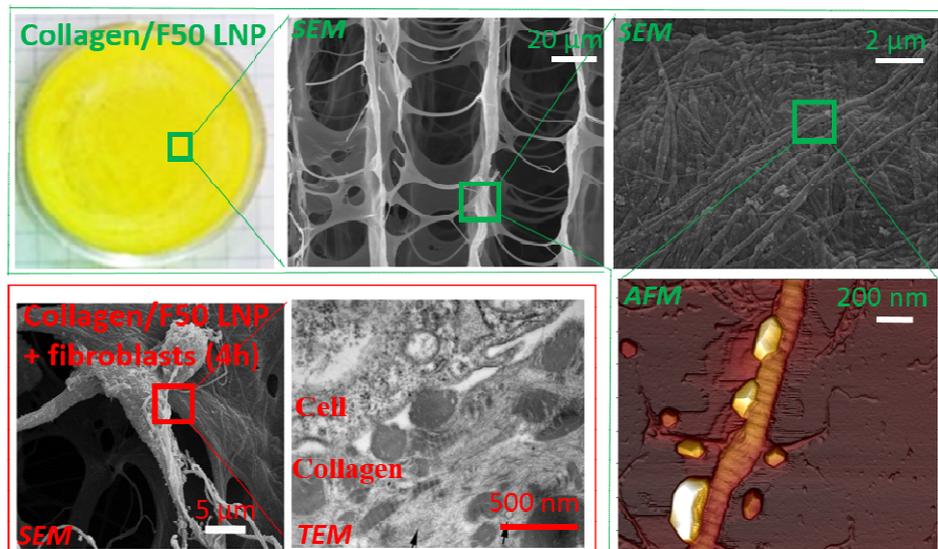
Abstract

Curcumin-loaded collagen cryostructures have been devised for wound healing applications. Curcumin displays strong antioxidant, antiseptic, and anti-inflammatory properties, while collagen is acknowledged for promoting cell adhesion, migration and differentiation. However, when curcumin was loaded directly into collagen hydrogels, it formed large molecular aggregates and clogged the matrix pores. A double encapsulation strategy was therefore developed by loading curcumin into lipid nanoparticles (LNP), and embedding these particles inside collagen scaffolds. The resulting collagen/LNP cryostructures have an optimal fibrous structure with $\approx 100 \mu\text{m}$ average pore size for sustaining cell migration. Results showed that collagen was structurally unaltered and that nanoparticles were homogeneously distributed amidst collagen fibers. Hydrogels soaked in saline buffer released about 20% to 30% of their nanoparticles content within 24h, while achieved 100 % release after 25 days. When exposed to NIH 3T3 fibroblasts, these hydrogels provided a satisfactory scaffold for cell interaction as early as 4 h after seeding with no cytotoxic counter effect. These positive features make the collagen/lipid cryostructures a promising material for further use in wound healing.

Keywords

Collagen ; curcumin ; lipid nanoparticles ; wound healing ; SEM ; TEM ; AFM

Graphical abstract



Short summary

Curcumin-loaded lipid nanoparticles (LNP) were loaded into collagen cryostructuresto design innovative wound healing materials. The collagen/LNP scaffolds were characterized by different techniques (SEM, TEM, AFM, DSC, compression...) to assess their structural and mechanical properties. They displayed good adhesion properties with both keratinocytes and fibroblasts, and double encapsulation strategy prolonged curcumin payload release.

Introduction

For a wound to heal successfully, several cellular and molecular mechanisms have to act in a proper temporal sequence.^[1-2] Improper healing causes the wound to enter an intransigent phase of chronicity, due to a high severity score of the injury itself or to a low health state of the patient. For instance, the fast growing incidence of diabetes mellitus expose many patients at the risk of developing such a chronic non-healing complication as foot ulcer.^[2] In these conditions, high concentrations of proteases, reactive oxygen species, senescent cells and impaired stem cells are all hallmarks of chronic wounds.^[1]

In recent years, artificial scaffoldings acting as skin substitutes have been developed to improve wound healing.^[1-2] Because of its relative abundance in bodily proteins, its structural role as a main component of the extracellular matrix, its low immunogenicity and high stability, collagen is *par excellence* the best biomaterial to be employed as a prosthetic medical device for wound healing, except for patients prone to keloid scarring, for whom collagen deposition in the wound bed is excessive.^[3-4] Collagen is made by the self-assembly and binding of tropocollagen molecules, composed of three helical polypeptide chains wound around each other to form a triple helix. The resulting fiber acts as a stable scaffold for cell attachment by means of numerous RGD integrin-binding motifs. Collagen is also known to play important roles in cell migration and differentiation, by counteracting elevated concentrations of metalloproteases, and eventually enhancing granulation tissues and neo-angiogenesis.^[5] Different collagen materials have been proposed for wound healing purposes, based on pure collagen^[5] or combined with other biocompatible polymers, such as alginate,^[6] chitosan,^[7] or cellulose.^[8] However, the adoption of aggressive extraction and processing procedures may denature collagen and modify its structural and chemotactic properties^[5]. In a previous study, a few commercial collagen-based materials were compared to evaluate how cells and matrix interact in relation to pore sizes, mechanical stability and susceptibility to

proteolytic degradation.^[9] The best structural conditions were provided by native collagen cryostructures obtained by freeze-drying of 1% w/v collagen gel in acetic acid, characterized by high matrix porosity and thick fiber bundles.^[9] Under these conditions, cells proliferated, migrated and adhered to the collagen fibers through extended filopodia.^[10-11]

Several therapeutic strategies have recently been proposed to complement these scaffold features with antimicrobial or wound-healing promoting properties^[12-13]. For instance, silver nanoparticles have been included in different commercial products (Acticoat®, Smith & Nephew; Altreet® Ag, Coloplast; Urgotuls Ag®, Urgo). In this study, we focused on the polyphenol curcumin which has been widely used for centuries in the traditional medicines of China and India. Curcumin exhibits some antimicrobial, antioxidant, antiseptic and anti-inflammatory properties, besides expressing a potential MMP inhibiting activity,^[14-15] and has recently aroused much interest for wound healing applications.^[16-17] It has also been demonstrated to accelerate wound healing via different biological processes, such as regulation of cytokine levels and MMP9 concentration in wound beds. Together, these factors enhance tissue granulation through fibroblast proliferation, help to deposit new collagen, stimulate neo-angiogenesis and regenerate epithelial layers.^[16-17]

New collagen/curcumin cryostructures were therefore designed with the intent of preserving all major hallmarks of collagen matrix for wound healing, i.e. high microporosity, convenient swelling ratio, good mechanical properties and favorable conditions for cell interaction. Since direct curcumin loading in the collagen matrix occluded the cryostructure pores, we show here that loading the drug in lipid nanoparticles (LNP) embedded in the cryostructure preserved all properties of the collagen scaffold.

Several physico-chemical analyses have also been carried out to check how properties of these cryostructures compared to those of pure collagen scaffolds. Structural interactions

with fibroblasts have also been assessed by Scanning Electron Microscopy (SEM) and Transmission Electron Microscopy (TEM).

2. Results and discussion

2.1. Preparation of curcumin-loaded collagen cryostructures

In this study, collagen type I extracted from equine Achilles tendon and dispersed in 1% w/v acetic acid at pH 3 was employed to obtain cryostructures by freeze-drying. Collagen cryostructures are known to absorb large amount of wound exudate and maintain a moist environment in the wound.^[3-5]

In a first experimental design, curcumin was dissolved in small volumes of ethanol and mixed with the collagen gel, and the resulting slurry gently stirred to ensure complete homogenization before freeze-drying (**Figure 1A**). However, when observed by SEM, the resulting matrix did not display the usual aspects of payload-free collagen, some pores appearing obstructed by curcumin aggregates (**Figure 1B**).^[10]

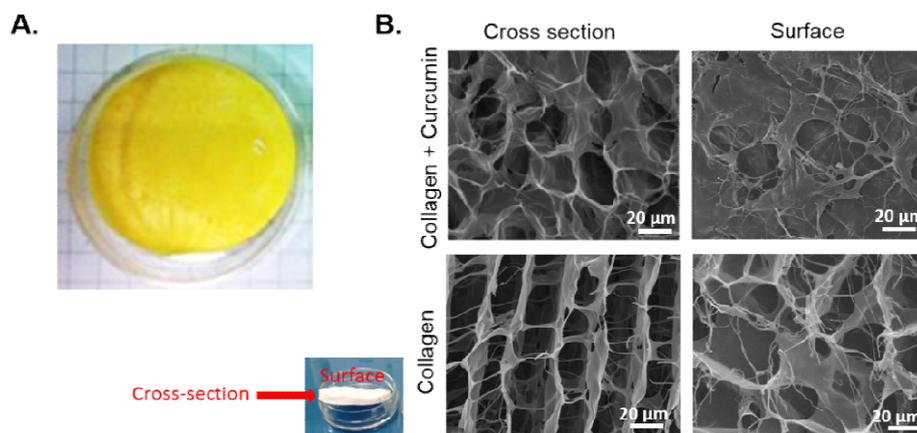


Figure 1. Collagen cryostructures prepared by direct curcumin loading (1 mg curcumin/g collagen). **A.** Photograph of the collagen scaffold. **B.** SEM images of collagen cryostructures loaded with curcumin (top) and compared with pure collagen cryostructures (bottom).

Curcumin is acknowledged for its low solubility and high chemical degradation rate in aqueous environment, along with low tissue absorption and short plasma life-time. These properties make its formulation for wound healing applications a challenging task.^[18] For loading curcumin in hydrogel scaffolds, several “double-encapsulation” strategies have been recently proposed. They all rely on the experimental formulation of curcumin-loaded polymer microparticles,^[8] nanoparticles,^[6] or nanomicelles,^[19-22] that are themselves encapsulated inside the hydrogel matrix. For instance, curcumin-loaded gelatin microspheres measuring a few micrometer in diameter were loaded in collagen-cellulose nanocrystal scaffolds,^[8] and 200 nm diameter drug-encapsulated chitosan nanoparticles were incorporated in collagen-alginate scaffolds.^[6]

In this study, curcumin was encapsulated in lipid nanoparticles (LNP). LNP are core-shell particles made of human-used approved ingredients, produced by up-scalable solvent-free high pressure homogenization. They can be used as stable non-toxic carriers for water-insoluble drugs or fluorescent molecular probes.^[23-24] These particles have been shown to act as drug reservoirs for the stratum corneum both in intact and damaged skin,^[25-26] and loaded with morphine or other opioids,^[27] astragaloside IV,^[28] antimicrobial peptides,^[29] or varying growth factors^[30] for wound healing purpose. Preliminary reports indicate that their lipid core may play positive roles in wound healing^[27, 31-33] by affecting the penetration rate of the encapsulated drugs^[26].

The double encapsulation technology developed in this study entails first curcumin entrapment into lipid nanoparticles followed by their uploading into collagen matrix (**Figure 2A**). Curcumin-loaded lipid nanoparticles with hydrodynamic diameters of 50 nm (F50) or 120 nm (F120) were synthesized using previously described protocols.^[34-35] Their main features are summarized in Figure 2B. The particle hydrodynamic diameters were adjusted by regulating the lipid and surfactant ratios.^[36] Encapsulation efficiency and final payload were

higher for F120 than for F50 (Figure 2B), due to their higher lipid/surfactant ratio and larger drug loading volume. F50 and F120 were then homogeneously dispersed in a collagen gel (collagen/lipid particle 10/1 w/w) by gentle stirring for about one hour, and the resulting slurry freeze-dried to produce a structurally uniform collagen matrix, as assessed by SEM (Figure 2C). The presence of nanoparticles of about 100 nm and 180 nm in diameter, for collagen/F50 and collagen/F120 samples respectively, could be envisioned by high resolution SEM (Figure 2C). At low magnification, collagen fibers are characterized by a trabecular pattern of vertically oriented laminae connected by thin filaments, as it is usual for collagen matrices.^[10] This organization remained invariant following nanoparticle addition (collagen/lipid ratio of 10/1 w/w), unlike encapsulation of chondroitin sulfate liposomes that was shown to cause an increase in collagen pore sizes in studies on osteoarthritis.^[37]

Pore sizes of the collagen/LNP cryostructures were measured directly by SEM using the ImageJ® software, and results are summarized in **Table 1**. All cryostructures, regardless of nanoparticle presence, displayed an average pore size of 80-100 μm , as previously reported.^[38-39] A pore size of about 50-150 μm is an optimal measure for sustaining cell migration in collagen matrices.^[38, 40] Collagen cryostructures with high porosity provide a healthy healing environment by facilitating diffusion of nutrients and oxygen, and absorbing exudate fluids and blood.^[10] Increase in lipid nanoparticles made the collagen matrix progressively more irregular, particularly if the collagen/lipid ratio is below 3.3/1 (Supporting Information Figure S1). The Feret's diameter was measured in collagen matrices differing in particle payloads. Data show that pore sizes remained invariant for up to very high collagen/lipid 2.5/1 ratio (Supporting Information Table S1), while started to increase for collagen/lipid ratios close to 2/1 (Table S1). This is in line with data from Atomic Force Microscopy (AFM) on hydrated samples showing that the collagen matrix became progressively more destructured for high

particle payloads (Supplementary Information, Figure S2). Therefore, unless otherwise specified, data presented in the following sections refer to cryostructures made with collagen/lipid ratios of 10/1 w/w.

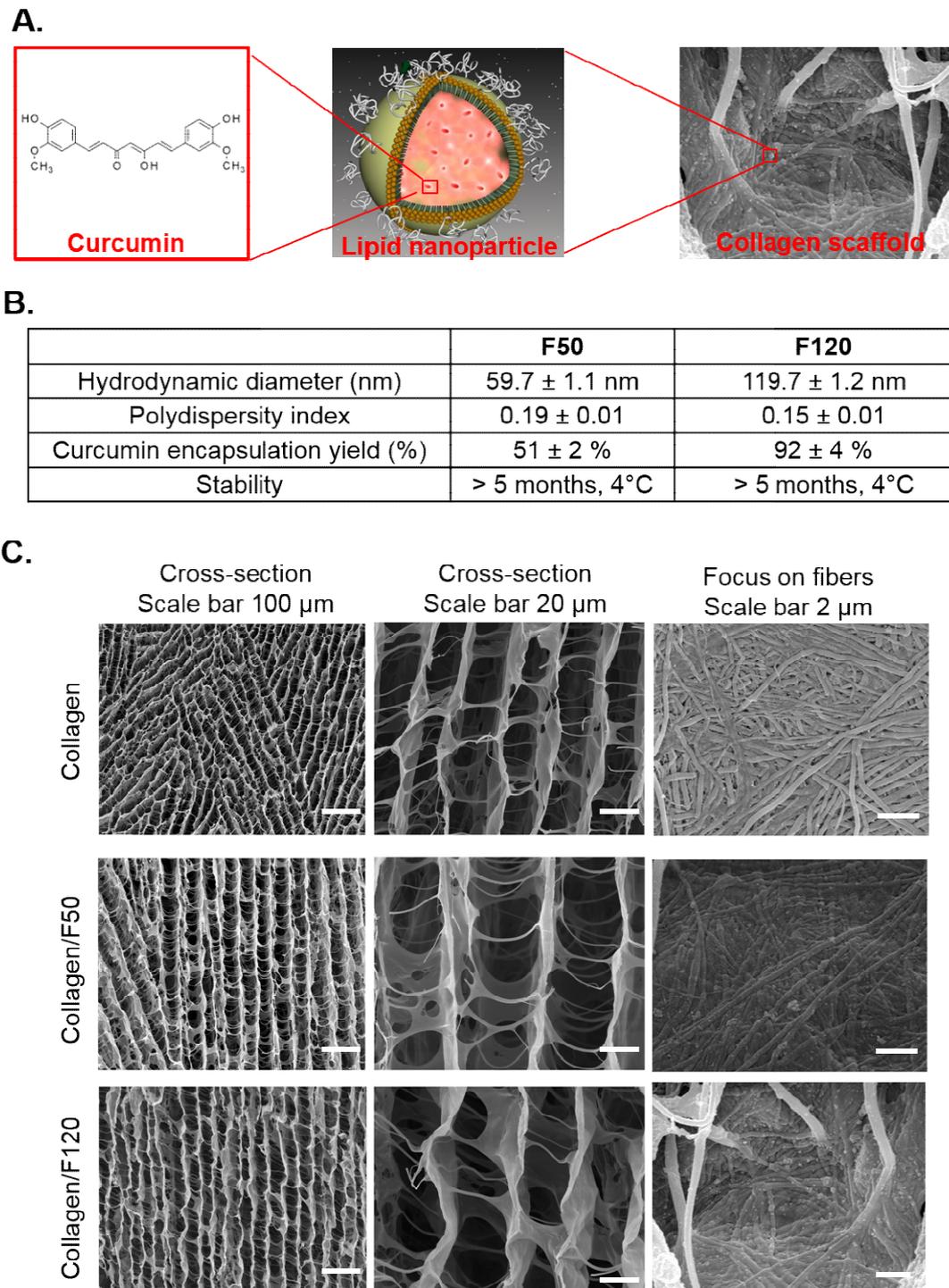


Figure 2. Cryostructures with doubly-encapsulated curcumin. **A.** Principle of double-encapsulating material. **B.** Properties of curcumin-loaded lipid nanoparticles. **C.** SEM images

of collagen versus collagen/lipid cryostructures for two different lipid nanoparticles (F50, F120).

2.2. Lipid particle distribution in cryostructures and material stability

To ascertain how lipid nanoparticles were distributed in collagen matrices, DiI and DiD fluorescent lipophilic tracers were loaded within LNP in a 1:4 ratio. LNP integrity and stability were then monitored by Fluorescence Resonance Energy Transfer (FRET).^[41]

FRET-labeled nanoparticles remained stable for about 24 hours in the collagen slurry, a time lapse sufficient for the subsequent gel freezing process (Supporting Information Figure S3). LNP distribution was then assessed by confocal laser scanning microscopy (CLSM) (Supporting Information, Figure S4). Under these conditions, DiI and DiD fluorescent signals appeared constant all over the collagen matrix, indicating that nanoparticles were uniformly dispersed and firmly embedded amongst the collagen fibers.

To assess the extent of particle stability in cryostructures in ambient atmosphere, nanoparticles were extracted from the collagen scaffold and examined at varying time-lapses following storage (**Figure 3A**). Both nanoparticle diameter (Figure 3B) and polydispersity index (Figure 3C) were measured by Dynamic Light Scattering (DLS). Data show that, when analyzed after collagen embedment and collagenase digestion ($t=0$), both F50 and F120 nanoparticles exhibited a mean size of about 170 nm and a polydispersity index around 0.25 (Figure 3B,C). These values were higher than those detectable in solution before LNP insertion (Figure 2B), but close to the ones exhibited after collagen gel mixing (Supporting Information, Figure S3). Nanoparticle sizes measured by SEM (Figure 2) did also indicate

larger values in cryostructures than in solution, a condition likely to depend upon the collagen absorption on the particle surface following gel digestion by collagenase.

However, particle size and polydispersity index remained invariant for the following 100 days (Figures 3B, 3C), indicating a remarkable particle stability. **Figure 3D** shows the particle FRET signal, resulting from the stability of DiI/DiD encapsulation and hence from the integrity of the particle core. The FRET signal was reduced by about $\approx 10\%$ and $\approx 40\%$ for F50 and F120, respectively, soon after cryostructure preparation ($t=0$). This indicates that some particles may have either leaked or disintegrated during preparation. The observation of a similar FRET signal decrease 24 hours after preparation (Supporting Information Figure S3D), makes it likely that leakage or disintegration occurred during gel mixing rather than during freeze-drying.

It has been previously shown that the dispersion of liposomes (lecithin shell) in a collagen gel enhanced their chemical stability but decreased their permeability.^[42] A similar stabilization effect was also observed by Shi et al^[43] and Mady^[44]. This effect was ascribed to the collagen capacity to interact with the liposomal surface and/or to the polymer insertion in the particle membrane.^[44] Likewise, the apparent phospholipid/surfactant shell destabilization of LNP could be attributed to the collagen presence. Thus, in spite of some loss during the first 10 days after gel preparation, the FRET signal remained stable for up to 23 days in the F120 sample and for 105 days in the F50 sample, indicating that a significant fraction of LNP in freeze-dried collagen scaffolds retained their DiI/DiD content during long term storage.

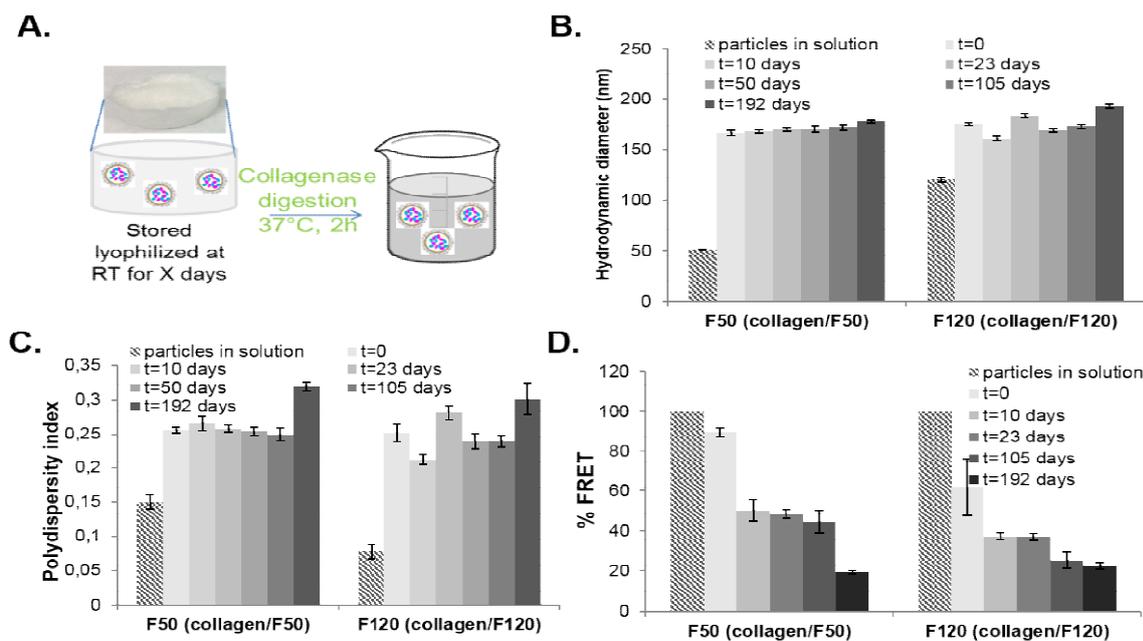


Figure 3. Stability of lipid nanoparticles embedded in cryostructures. To monitor their stability during ambient storage, particles were extracted from collagen cryostructures by collagenase digestion (A), and characterized by DLS (hydrodynamic diameter (B) and polydispersity index (C)) and measurement of FRET ratio (D).

2.3 Structure of collagen fibers

Native collagen is an organized structure comprised of tightly interwoven fibers resulting from the self-assembly of tropocollagen (3 peptide chains arranged in a triple helix conformation). Due to the staggered arrangement of the constituting polypeptides, collagen fibers exhibit a distinctive banding pattern (D-band, spaced about 67 nm).^[45] Collagen can be structurally altered by either chemical cross-linking or biomedical handling. An example is provided by the introduction of carbon nanotubes in collagen gels that elongated the D-band pattern.^[46] Given this possibility, the ultrastructure of collagen/lipid nanoparticle cryostructures was further explored by TEM (Figure 4) and AFM (Figure 5). Under these conditions, collagen cryostructures appeared comprised of filamentous and laminar structures made of compact fibrils exhibiting the typical banding patterns of native collagens. At the TEM level, nanoparticles appeared as spherical structures uniformly dispersed amidst collagen fibers (Figures 4 A,B). No nanoparticles could be envisioned in control

collagen gels (Figure 4C) as well as in gels mixed with the MyrjTMs40 co-surfactant prior to freeze-drying (Figure 4D). The expected range of the nanoparticle sizes (50 nm or 120 nm diameter, Figure 4) was also confirmed by image analysis. However, when observed by AFM, nanoparticles appeared as faceted nanostructures in the form of aggregates or attached to collagen fibers, and slightly larger than lipid particles (Figure 5A, 5B).

The particle faceted aspect and their size increase could be due to a process of lipid crystallization, even though this could not be demonstrated by DSC analysis in aqueous solutions.^[36] Lipid crystallization could also be due to LNP interacting with collagen fibers or, more probably, to the spin coating of the collagen/LNP film on silicon substrate. Interestingly, D-band spacing pattern of collagen/LNP samples was not significantly modified, indicating that collagen fibers were well preserved in the presence of nanoparticles (Figure 5D). The overall structural organization of these collagen fibers could be maintained for nanoparticle concentration up to a collagen/lipid ratio of 2/1 (Supporting Information Figure S5), while it became more relaxed for collagen/lipid ratios below 2.5/1.

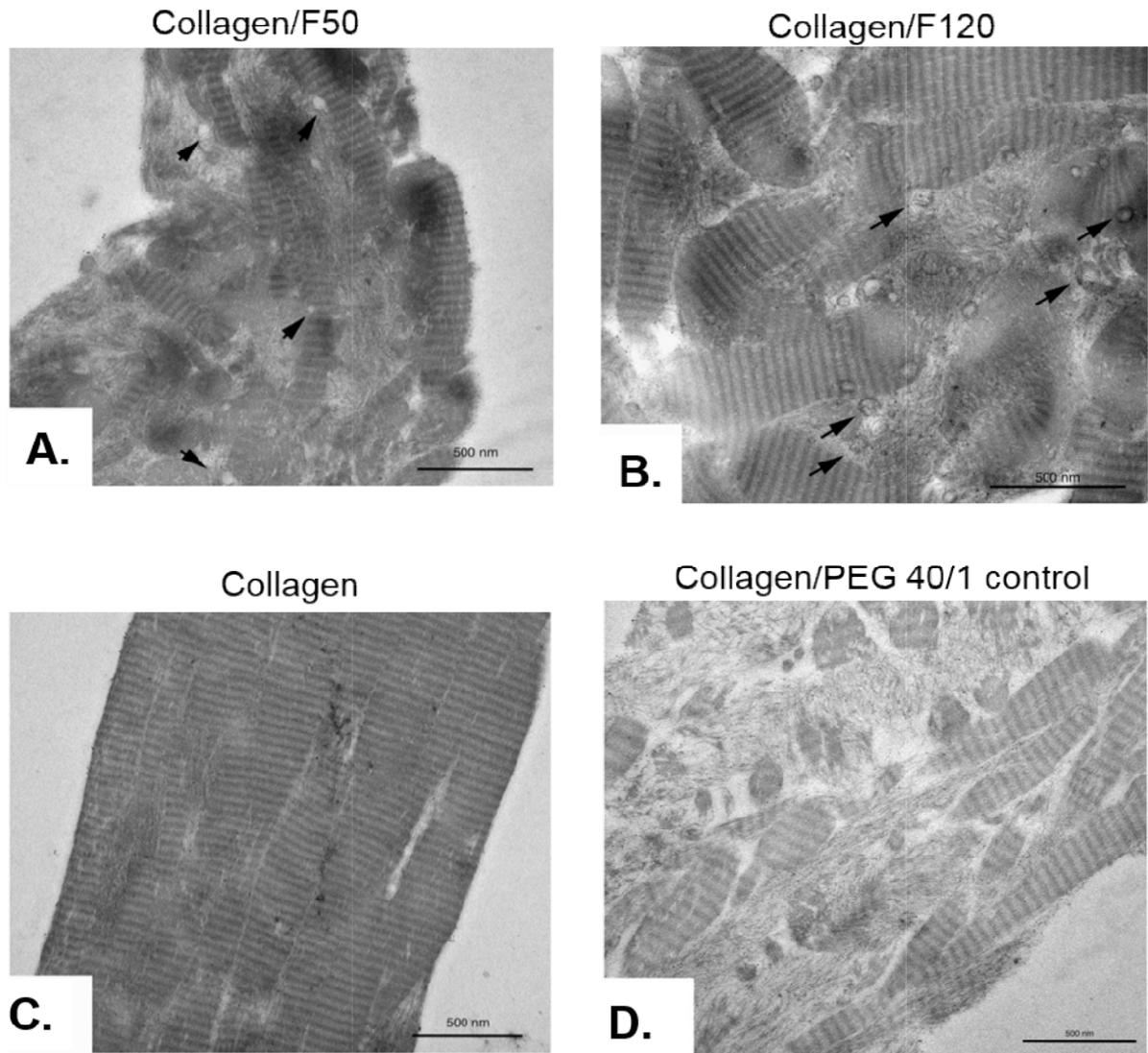


Figure 4. TEM analysis of sections of collagen cryostructures(scale bars: 500 nm).

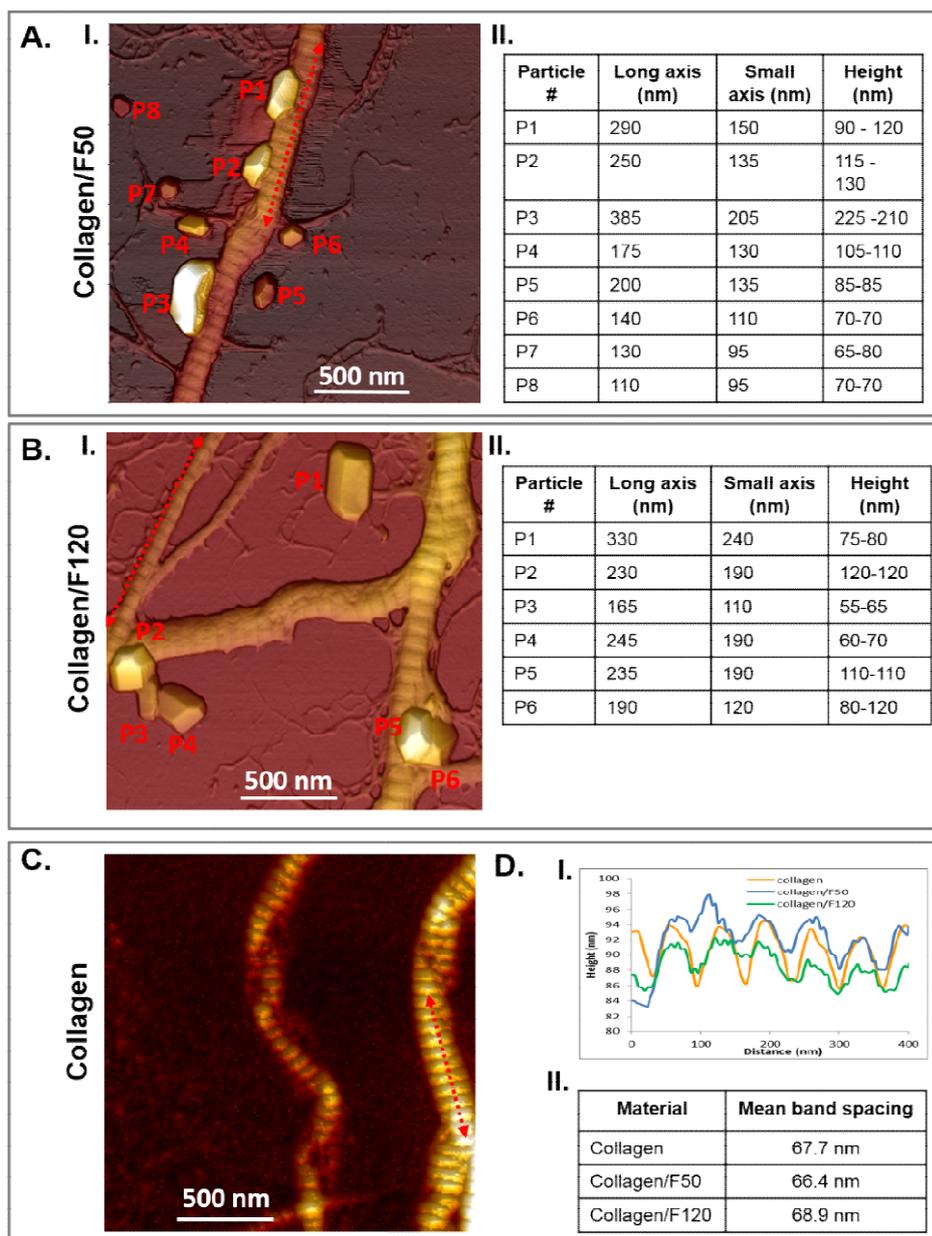


Figure 5. AFM analysis (in air) of collagen (C) or collagen/lipid nanoparticle films (A, B : I. images; II. Particle size analysis). D. Band spacing analysis of the collagen fibers: I: height profile along the arrows indicated in images (AI, BI, C) and II: mean band spacing measured for different samples.

The observation that, under these conditions, the collagen ultrastructure was preserved was confirmed by determining its thermal stability with Differential Scanning Calorimetry (DSC).^[47] Data are summarized in Table 1 and plots are displayed in Figure S6 of Supporting Information. The shrinkage temperature was determined as the onset value of the endothermic peak and the thermal denaturation temperature (for helix-to-coil transition of the collagen

fibers) was taken at the endothermic peak maximum. The value of denaturation enthalpy was calculated with respect to the weight of dried collagen. Particle coating did not affect the collagen structure, as indicated by the observation that it did not impact the thermal denaturation temperature of both collagen/F50 and collagen/F120 samples (Table 1). However, particle coating augmented significantly its shrinkage temperature ($>+3^{\circ}\text{C}$), suggesting a potential protective role. Higher denaturation enthalpies were also observed for collagen/F50 and collagen/F120 samples, in line with the expectation that the fiber helical structure was protected when coated with lipid nanoparticles. The thermal stabilizing effect of LNP was, however, lower than that measured in cross-linked collagen.^[6, 47]

Table 1. Physico-chemical parameters of the collagen cryostructures.

Material	Structural characterization (SEM)	Thermal Characterization (DSC)			Mechanical characterization E* [kPa]	Swelling ratio (SR)
		Shrinkage temperature (onset) [$^{\circ}\text{C}$]	Thermal denaturation temperature (peak) [$^{\circ}\text{C}$]	Denaturation enthalpy [J/g]		
Collagen	87 ± 58	49.8	59.3	42.9	111.4 ± 6.2	5.28 ± 0.16
Collagen/F50	99 ± 62	53.6	60.1	53.2	73.3 ± 31.0	5.22 ± 0.22
Collagen/F120	98 ± 60	55.3	59.3	58.4	57.1 ± 3.0	5.02 ± 0.25

2.4. Mechanical properties, swelling ratio and particle release

Collagen/lipid cryostructures appeared qualitatively softer (i.e. more deformable under compression) than pure collagen materials. Compression experiments were therefore performed to assess the mechanical properties of the cryostructures and stress-strain curves are displayed in Supporting Information Figure S7. The stress-strain plots obtained were typical of elastic foam-like materials,^[48] and the elastic moduli E* of the scaffolds were calculated as the slopes of stress-strain curves for very low strains (linear domain,

deformation < 3%) (Table 1). E^* were in the range of previously reported values for non-crosslinked equine collagen I scaffolds,^[47-48] and were indeed lower for the collagen/LNP scaffolds in comparison to pure collagen cryostructures, with a more pronounced effect for collagen/F120 than for collagen/F50. This is indicative that LNP did not act as rigidifying matrix cross-links as observed for other nanoparticle types, such as gold nanoparticles^[49] or cellulose nanocrystals.^[50]

The gel absorbing capacity was also quantified by measuring its swelling ratio in 1X PBS (10 mM phosphate, 0.137 M NaCl, pH 7.4). Data show that, in line with other observations,^[47] collagen gels took up ≈ 5 folds their weight of water (Table 1). However, slightly lower swelling ratios could be observed in the presence of lipid nanoparticles, and more so for the F120 than for the F50 sample, perhaps in response to the hydrophobic effect exerted by nanoparticle coating. Overall, these cryostructures exhibited very good absorbing properties under conditions that are typical of the biological fluids released during wound healing. These swelling properties, along with the cryostructure microporosity, should favor nanoparticle diffusion outside the scaffolds. However, the double encapsulation strategy has been shown to prolong the kinetics of drug release from the collagen hydrogels. For instance, gentamicin was released in about 10 days from collagen/PLGA microparticle hybrid cryostructures, while the free drug diffused out of collagen material in 2 hours.^[12]

Given these alternatives, the release of DiI/DiD fluorescent dyes (i.e. in or out the lipid nanoparticles) from collagen/LNP scaffolds was studied for collagen/F50 and collagen/F120 cryostructures using fluorescence and FRET quantification (**Figure 6A**). Results show that both dyes diffused out of the collagen cryostructures quite slowly, displaying two different phases. In the first “burst release” phase, about 30% of the F50-loaded dye and 20% of the F120-loaded dye were released in 2 days. As indicated by DLS and FRET measurement on gel supernatants, at this stage LNP were still intact and unchanged in

sizes (Supporting Information Figure S8). In a second “prolonged release” phase, dyes escaped more slowly and achieved a complete release within 25 days. It thus appears that, in the time lapse from day 14 to 25, dyes were released partly in the form of encapsulated particles and partly as free molecules (Supporting Information Figure S8). Similar diffusion patterns have been reported for other lipid nanoparticle/hydrogels (in particular polysaccharide-based) by us^[51] and others^[52].

Lipid nanoparticle diffusion from hydrogels is a diffusion-driven process.^[52] The first “burst release” phase could be accounted for by the fast diffusion of LNP directly contacting the water phase, whereas deeper particles would take longer to escape.^[51] Apparently, diffusion is more influenced by charge and hydrogel pore size than by particle size.^[51-52] Indeed, as observed here, the diffusion kinetics of F50 was only a bit faster than that of F120 nanoparticles (Figure 6A).

The study of drug/particle diffusion in PBS buffer provides good comparative indications for understanding the kinetics of drug delivery. However, the wound exudate is known to contain proteases that induce drug release through collagen digestion. Since native collagen is known to be quickly degraded in the presence of bodily fluids, the adoption of cross-linking strategies is generally pursued to prolong its lifetime.^[4] The kinetics of collagen degradation was thus assessed by titrating the 4-hydroxyproline content in the presence of collagenase (1U/mg) (**Figure 6B**). Under these conditions, three days of incubation sufficed to degrade collagen completely, native collagen and collagen/lipid nanoparticle cryostructures exhibiting similar degradation kinetics. In this experiment, the presence of lipid nanoparticles did not prevent collagen degradation, while chondroitin sulfate-loaded liposomes have been previously reported to decrease collagen biodegradability in hydrogels.^[37] This difference may be accounted for by the steric hindrance exerted by liposomes at collagenase cleavage sites. However, liposome loading decreased significantly the porosity, while increased pore

size, of the collagen scaffold ^[37](Figure 2, Table 1). These latter considerations make collagen interactions with the LNP quite different from those observed with liposomes. Overall, collagen/LNP cryostructures could be reasonably counted amongst the most efficient wound dressings, displaying such remarkable features as high mechanical strength, high porosity, satisfactory swelling ratios and a prolonged drug release.

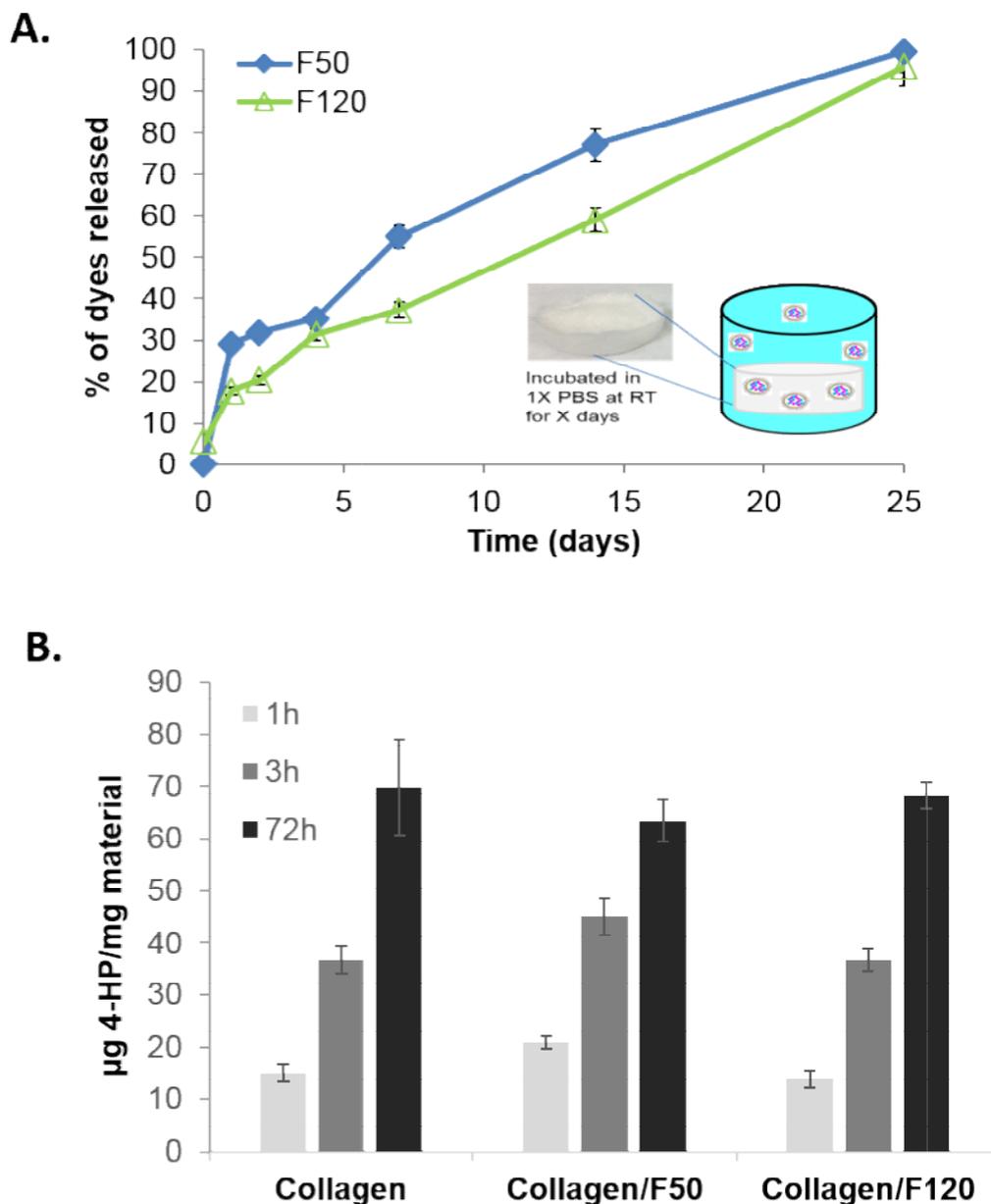


Figure 6. Kinetics of dye release from collagen/lipid cryostructures immersed in 1X PBS buffer(A) and collagen biodegradability in presence of collagenase (B).

2.5 Cell proliferation and interaction with cryostructures

Collagen is known to play key roles in cell adhesion, migration and differentiation, due to the presence of RGD integrin recognition sequences.^[4] Therefore, to verify how collagen/LNP cryostructures accomplish these functions their interactions with fibroblasts and keratinocytes was studied *in vitro*. To this end, NIH 3T3 fibroblasts were seeded on cryostructures (collagen, collagen/F50, collagen/F120) contained in 35 mm Petri dish in a 2×10^5 cells/matrix ratio. Cell proliferation was monitored by measuring the mitochondrial enzyme succinate dehydrogenase activity (MTT assay). MTT activity started to increase 48 hours after seeding both in collagen and cryostructures loaded with 10% w/w lipid nanoparticles (Supporting Information **Figure S9**). SEM and TEM images of cryostructures exposed to NIH 3T3 fibroblasts for 4 hours or 3 days are presented in **Figure 7**. Four hours after seeding (Figure 7A), cells appeared firmly anchored on both collagen and collagen/LNP cryostructures through the extension of several filopodia (Figure 7 A.II). At the ultrastructural level, several contact points could be envisioned between fibroblasts and collagen/F50 fibers (Figure 7A.V), which also included intact 50 nm particles (arrows) adhering onto some collagen fibers and other fibrillar material (Figure 7A.V). Figure 7A.VI shows a NIH 3T3 fibroblast of the collagen/F120 sample with an incorporated collagen fiber.

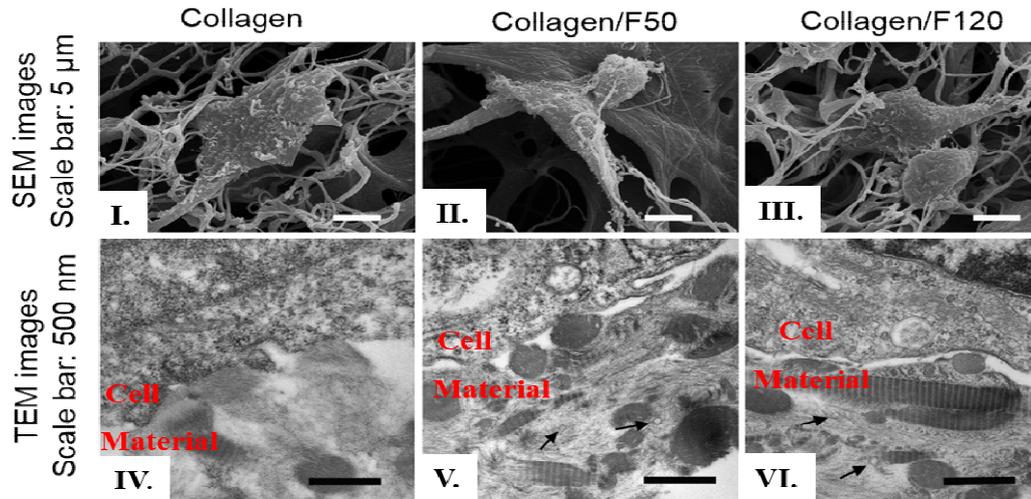
The extent of cell interaction increased with prolonged culturing times (Figure 7B), conferring a more relaxed aspect to the collagen fibres (Figures 7B.I and IV). By comparison, fibres of the collagen/F50 (Figures 7B.II and V) and collagen/F120 (Figures 7B.III and VI) samples remained more compact, making visible some of the 50 nm (Figure 7B.V) or 120 nm (Figure 7B.VI) nanoparticles. At this incubation time, NIH 3T3 fibroblasts of the collagen/F50 sample appeared to have incorporated some collagen fibres (Figure 7B Figure

7C). Figure 7C-I shows cytoplasmic protrusions enveloping collagen fibres and F50 nanoparticles, while Figure 7C-II shows a nanoparticle in contact with the cell membrane.

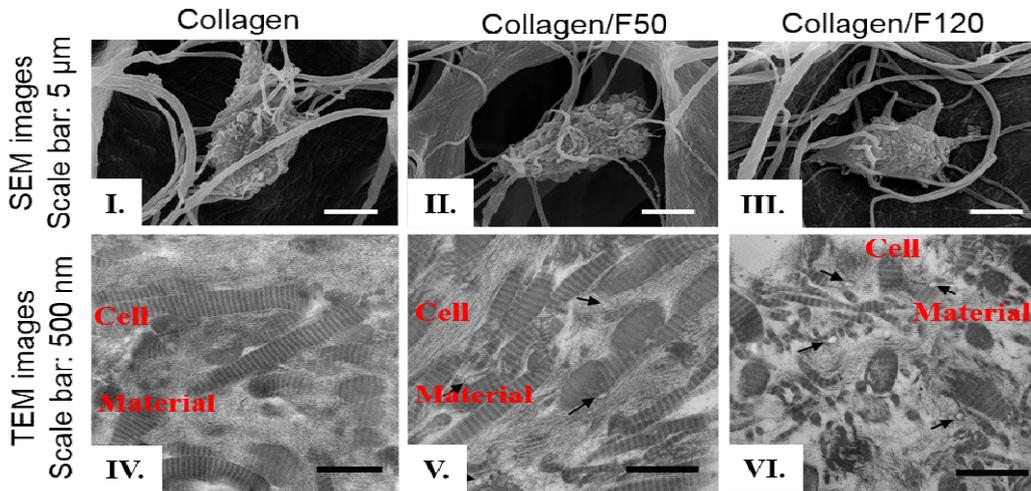
Similar experiments were conducted using HaCat keratinocytes seeded on collagen cryostructures at the 2×10^5 cells/matrix ratio. Three days of *in vitro* culturing sufficed to show that cells proliferate quite well regardless of the collagen matrices (Supporting Information, Figure S9). However, while collagen/F50 and collagen/F120 samples could still exhibit some intact fibers after either 3 days (Figure S10) or 7 days (Figure S11), control collagen matrix was completely degraded.

In conclusion, both collagen/F50 and collagen/F120 cryostructures displayed good adhesion properties for both keratinocytes and fibroblasts. Apparently, the porosity and mechanical strength provided by these scaffolds were suitable for cells to migrate and adhere on the collagen matrix. As well known, native collagen is a MMP substrate. In the absence of any MMP-inhibiting activity, as in certain chronic wounds, the metalloproteinase level is not properly regulated,^[4] and collagen is consequently quickly degraded. As a result, collagen peptide fragments lose any binding activity for sequestering inflammatory mediators.^[3-4] In the attempt to overcome this difficulty cross-linked collagens have been used, though soon substituted by native collagen when applied onto a wound.^[3] In the present study, we proved that fibres of the collagen/LNP cryostructures were more resistant to enzymatic degradation than pure collagen (Figures 7 and S10-S11). The observation that LNP maintained a fibrillary structure in collagen cryostructures could be of high interest for designing slow degrading non-cross-linked scaffolds for prolonged application at the wound site. In addition, the observation that fibroblasts and keratinocytes incorporated collagen suggest the possibility that the associated nanoparticles could be used as potential tools delivering specific payloads to the cells involved in wound healing (Figure 7C).

A. 4 h



B. 72 h



C. 72 h (focus)

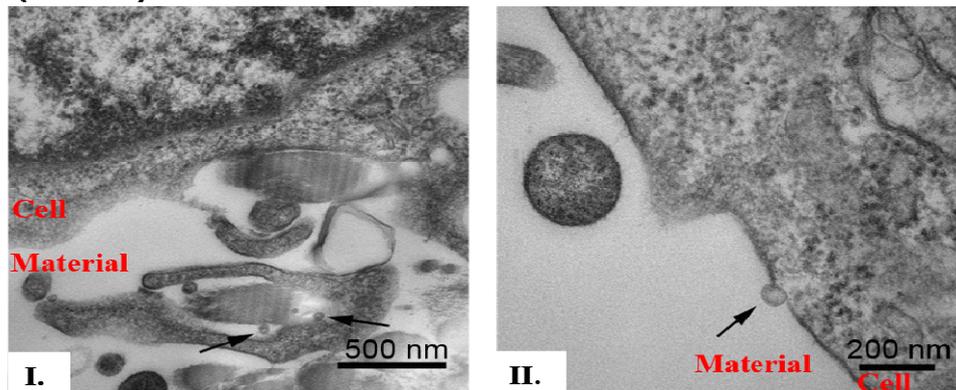


Figure 7. NIH 3T3 fibroblast/collagen cryostructure interactions assessed by SEM and TEM 4 hours (A) and 3 days (B,C) after cell seeding.(C) TEM images focusing on points of interaction between a NIH 3T3 cell and collagen/F50 cryostructures 3 days after seeding. Nanoparticles are indicated by arrows.

3. Conclusion

The ideal biomaterial promoting wound healing should display different features.^[4] First, it should attract cells capable of restoring new tissues. Collagen has been shown to be a biomaterial of choice to achieve this goal for its high biocompatibility, non-immunogenicity and for expressing a large number of RGD integrin-binding sites. Second, it should promote cell proliferation, modulate proteolytic activity and guide the organization of new ECM. Chronic wounds develop whenever MMPs overexpression results in excessive matrix degradation.^[3-4]

In this study, we demonstrated that curcumin-loaded collagen cryostructures came to express these desirable features when recast by double encapsulation technology. Curcumin-loaded nanoparticles did not significantly modify such structural properties as high porosity, swelling ratio, matrix degradability and cell interactions, indicating that they entertained only feeble interactions and no cross-links with the collagen fibers. Even though nanoparticles entrapment made the collagen cryostructures softer than controls, cryostructures proved capable of maintaining excellent integrity when soaked in buffered saline. Thanks to the double encapsulation strategy, curcumin was released rather slowly from the scaffold, with the potential benefit for a prolonged therapeutic effect. There were also preliminary indications that fiber coating by LNP could slow down their degradation rate and prolong their structural persistence. Overall, curcumin-loaded collagen/lipid nanoparticle cryostructures promised to provide innovative scaffolds for treating chronic wounds.

4. Experimental Section

Materials: Gel of 1% w/w collagen type I extracted from equine Achilles tendon in acetic acid pH 3 was kindly provided by Euroresearch. Briefly, tendons were removed of all surrounding tissues, harvested and cleaned in deionized water. They were then treated with

10% sodium chloride at 4°C for 24 h under continuous magnetic stirring. The mixture was centrifuged and the precipitated collagen fibers repeatedly washed and eventually recovered with a 0.5% w/v solution of acetic acid.^[10] Curcumin (> 97%) was purchased from TCI Europe N.V. (Paris, France), Suppocire NB™ from Gattefossé (Saint-Priest, France), Lipoid S75 (soybean lecithin at >75% phosphatidylcholine) from Lipoid (Ludwigshafen, Germany), Myrj™ S40 (polyethylene glycol 40 stearate) and super refined soybean oil from CrodaUniqema (Chocques, France), 1,1'-Dioctadecyl-3,3,3',3'-tetramethylindocarbocyanine perchlorate (DiI) and 1,1'-dioctadecyl-3,3,3',3'-tetramethylindotricarbocyanine perchlorate (DiD) from LifeTechnologies (Saint-Aubin, France). Other chemicals were purchased from Sigma-Aldrich (Saint-Quentin-Fallavier, France).

Nanoparticle synthesis and characterization: Lipid nanoparticles of different diameters (50 nm: F50, 120 nm: F120) were prepared according to previously described protocols.^[34] Briefly, an oily mixture composed of soybean oil (85 mg/150 mg, respectively for F50/F120), Suppocire NB™ (255 mg/450 mg respectively) and Lipoid-s75® (65 mg/45mg respectively), added with curcumin (7.5 mg/8.6 mg respectively) or dyes (DiI or/and DiD), supplied in ethanol, was mixed with an aqueous phase composed of Myrj™ S40 (stearate-PEG₄₀, 345 mg/215 mg respectively) in 10 mM phosphate buffer pH 7.4 (1X PBS, 1250 µL/1140 µL respectively). Sonication cycles were performed at 45°C for 5 minutes to yield particle dispersion with the desired hydrodynamic diameter. LNP were protected from light and dialyzed overnight in 1X PBS (10 mM phosphate, 0.137 M NaCl, pH 7.4) at room temperature. Samples were diluted to obtain a concentration of 100 mg/ml of lipids (including soybean oil, Suppocire NB™, lecithin and Myrj™ S40). LNP were then sterilized by filtration using 0.22 µm filters, stored at 4°C and light proof. Dynamic light scattering (DLS) was used to determine particle hydrodynamic diameter and zeta potential (ZetaSizer Nano ZS,

Malvern Instrument). Particles were diluted to 2 mg/ml of lipids in sterile 0.1X PBS (1 mM phosphate, 13.7 mM NaCl, pH 7.4) and transferred in ZetaSizer Nano cells (Malvern Instrument) before each measurement, performed in triplicate. Curcumin encapsulation yield was determined by HPLC analysis after particle destabilization in acetonitrile. Chromatographic analysis was achieved on a C18 column (Sunfire™, 5 mm, 250 × 4.6 mm, 100 Å) using an Alliance 2695 separation module equipped with a controller 600 and a 2487 Dual Absorbance Detector set at 265 and 420 nm (Waters, USA), and data were processed using Empower 2 software. The elution method used a binary gradient mobile phase consisting of (A) a mixture of water acidified with 2% (v/v) acetic acid and (B) acetonitrile, at a 1 ml/min flow rate. The gradient was applied as following: 0–12 min from 60 to 100% B; 12–17 min from 100% to 60% B; 17–20 min 60% B.

Collagen cryostructure preparation: 25 g of collagen gel at 1% w/w in acetic acid buffer (containing 250 mg of collagen) were mixed with the appropriate volume of F50 or F120 at 100 mg/ml for 1 hour at room temperature (20°C). 2 g of gel material were distributed into each well of 12 well cell culture plates. The upper surface was covered with a polypropylene membrane before freezing in liquid nitrogen for 5 min. The freeze-drying process was done for 24 hours with a Cryotec V8.11 plate lyophilizer. Control collagen/Myrj™ s40 (collagen/PEG) materials were prepared at the 40/1 w/w ratio. Collagen cryostructures directly loaded with curcumin were prepared using the same procedure by mixing 25 g of collagen gel with 250 µg of curcumin dissolved in 250 µl of ethanol.

Particle stability when embedded in the collagen cryostructures: Nanoparticles were extracted from cryostructure to determine their stability during storage. 1 ml of collagenase

solution (1.5 mg/ml collagenase, 1 mg/ml CaCl₂ in 1X PBS) was added to 4 mg of lyophilized collagen and incubated at 37°C for 3 hours. Undigested materials, precipitated calcium chloride and collagenase were eliminated by centrifugation at 10,000 g for 2 minutes. The supernatant, containing digested collagen and nanoparticles, was analyzed by DLS measurements (particle colloidal stability) and by fluorescence spectroscopy using a Perkin-Elmer LS50B fluorimeter (DiI and DiD co-localization reflects the LNP integrity, as assessed by the DiD/DiI fluorescence ratio^[41]). FRET data were expressed as % of FRET, calculated as: % FRET = 100 × [(DiD)/DiI (t) - DiD/DiI (100% dissociation)] / (DiD/DiI (fresh solution) - DiD/DiI (100% dissociation)]. The DiD/DiI fluorescence ratio at 100% dissociation was obtained by incubating nanoparticles in DMSO to yield to their total destruction in a few minutes.

Scanning Electron Microscopy: Collagen scaffolds were cut with a sterile razor blade into a number of small blocks and attached to aluminum stubs with a carbon tape. Blocks were oriented to expose either the upper or the lower surface, while others were glued on their longitudinal cut profile. For SEM analysis, collagen/lipid particle cryostructures and their cellular loads were cut into a number of small blocks with a sterile razor blade and fixed overnight with 3% glutaraldehyde at 4 °C in phosphate buffer (PB) at pH 7.2. Following extensive rinsing with the same buffer at 4 °C, they were immersed for 1h in 0.5% tannic acid in PB at 4 °C, rinsed again four times in the same buffer for 15 min at 4 °C, and eventually post-fixed for 1h with 1% osmium tetroxide in PB at 4 °C. Specimens were finally washed in distilled water, block-stained with 1% uranyl acetate in distilled water and then dehydrated in a graded series of ethanols. Samples were first dried by the critical point method in a Balzers Union CPD 020 and then sputter-coated with gold in a Balzers MED 010 unit to be observed in a JEOL JSM 6010LA electron microscope.

Transmission Electron Microscopy: Collagen/lipid particle cryostructures were cut into a number of small blocks with a sterile razor blade and fixed as specified above. Following dehydration in a graded series of ethanol, samples were infiltrated for two days in graded mixtures of LRWhite resin/ethanol. By the end of this procedure, samples were embedded in fresh LRWhite resin and cut with Reichert Ultracut ultramicrotome equipped with a diamond knife. Ultrathin sections (60-80 nm thick) were collected on copper grids, stained with uranyl acetate and lead citrate, and observed with a JEOL 1200 EX II electron microscope. Micrographs were acquired by the Olympus SIS VELETA CCD camera equipped with the iTEM software.

Atomic Force Microscopy: Collagen/LNP films were deposited and dried on SiO₂ substrates by spin coating for AFM observation in air. We used PeakForce Tapping on a Dimension ICON microscope (Bruker) with a “ScanAsyst Air” tip (Bruker). The stiffness of each tip ($k \sim 0.4$ N/m) was obtained by thermal noise method after photodiode calibration on a hard and rigid substrate (Al₂O₃). Images were acquired with an average setpoint force of ~ 500 pN. Scan speed was set and adapted during imaging to obtain an average error on force feedback less or equal to 10% of the force setpoint value.

Differential Scanning Calorimetry: 33-45 mg dry collagen were soaked for 1 h in deionized water, dried on filter paper for 1 minute, weighed again and sealed in aluminum pans. DSC analysis was carried out on a micro DSC III SETARAM calorimeter. An aluminum pan with the same weight of deionized water was used as a reference. Heating rate was set at a constant value of 3°C/min and the temperature increased from 20 °C to 110°C. Several duplicates (2 to 5) of the same sample (from different batches) were analyzed.

Uniaxial compression test: Uniaxial compression tests were performed on lyophilized cylinder shaped collagen of 4.1 mm diameter and 5 mm height using an AR2000EX rheometer (TA instrument) with an initial preload of 0.03 N and a compression speed of 10 $\mu\text{m/s}$. Stress σ was calculated as the normal force divided by the collagen surface area, whereas strain ϵ was expressed as the percentage of collagen compression.

Swelling ratio: Collagen scaffolds were cut in small pieces (about 1 cm x 1 cm), weighed and soaked for 24 hours in 5 ml of phosphate-buffered saline (PBS) buffer, at room temperature (20°C). Non-absorbed solution was removed by aspiration and wet materials dried for 3 minutes on a filter paper. Finally, wet collagen samples were weighed and their swelling ratio (SR) calculated according to the following equation: $\text{SR} = (W_w - W_d) / W_d$ in which W_d is the dry weight (i.e. lyophilized collagen prior to PBS soaking) and W_w is the resulting wet weight. 3 to 6 sample /point were used.

Kinetics of release of the dyes from the collagen cryostructures: For each time point, 5-10 mg samples were incubated at 37°C in 1 ml 1X PBS. 300 μl of supernatant were then taken up and analyzed by DLS measurements (particle colloidal stability) and fluorescence spectroscopy using a Perkin-Elmer LS50B fluorimeter. Dye concentration and FRET percentage were calculated from the fluorescence spectra, as specified above.

Kinetics of degradation of the collagen cryostructures: Kinetics of the cryostructured degradation was measured by quantifying the 4-hydroxyproline (Hyp) released following collagenase (*Clostridium histolyticum*) digestion. Five mg of each sample were incubated at 37°C with 5 units of collagenase solution (Sigma-Aldrich C0130) diluted to 5

ml of 0.01M Tris-HCl, pH 7.2, containing 50mM CaCl₂, for the desired time of digestion. The solution was filtered with a 5µm pore polypropylene filter to remove the degradation products. 4-Hyp concentration was determined using the Hydroxyproline Assay Kit (Sigma-Aldrich, MAK008).^[53]

Cell culture: Human keratinocytes (HaCat) and stabilized murine fibroblasts (NIH 3T3) were cultured in 75 cm² flasks containing DMEM (Dulbecco's Modified Eagle's Medium), supplemented with 10% of newborn calf serum (NCS) and fetal bovine serum (FBS) respectively, 1% glutamine and 1% penicillin/streptomycin, and maintained in an incubator at 37 °C with 5% CO₂ and controlled humidity (cell incubation time 24 h). Upon reaching confluence, cells were passed into new culture vessels in a 1:5 ratio and the medium changed every three days.

Cell proliferation: Cell proliferation was assessed using the MTT assay. NIH3T3 mouse fibroblasts (2x10⁵ cells/well) were seeded in 12-well plates containing glycosylated collagen (25 mg) and incubated for 24, 48 and 72 hours with 2 ml of culture medium. At each incubation time the culture medium were withdrawn and 0.2 ml MTT (Sigma-Aldrich) were added to every aliquot and left to incubate for 4 hours in the dark at 37 °C under a 5% CO₂ flow. By the end of this incubation time, the resulting formazan crystals were completely dissolved in DMSO and the relative absorbance measured at a wavelength of 570 nm. Cells cultured with no collagen and cells incubated in presence of zinc were used as positive and negative controls, respectively. The percentage of cell proliferation was determined using the following equation: Cell proliferation (%) = ((AS-ANC)/(APC-ANC) × 100) where AS, represent the sample absorbance; ANC, the negative control (cells with zinc material); and APC, the positive control (only cells).

Collagen/Cell interaction: Cells were seeded on the top surface of lyophilized collagen scaffolds in 35 mm Petri dish, with 2×10^5 cells/matrix. Collagens and cells were then incubated at 37°C in 2 ml of culture medium under a continuous flow of 5% CO₂. Medium was then renewed every 3 days. SEM and TEM imaging were performed as described above.

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